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A facile route for hydroxyapatite densification with an increased heating rate



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ABSTRACT

The densification behavior of hydroxyapatite nanorods prepared by chemical precipitation method in open reactor conditions was investigated by application of different heating rates. The non-isothermal processing was performed with 2, 10, and 50 °C/min up to 1200 °C and yielded fully dense ceramics. The implementation of the higher heating rate provided grain size refinement from micrometer level for the slowest ramp, down to 250 nm in the case of processing with 50 °C/min, without any drawbacks regarding final density. The relative amount of retained structural hydroxyl groups in sintered ceramics was gradually increased with the heating rate. Furthermore, the qualitative level of optical translucency was increased with a higher heating rate which can be explained by the beneficial alignment of HAp nanorods during the fast heating rate processing, achieved microstructural refinement, and higher amount of structural hydroxyl groups.

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1. Introduction

Conventional sintering of HAp was usually referred from the standpoint of the sintering temperature, while less attention was given to the heating rate, especially in the case of conventional pressureless sintering. However, with progress in synthesis procedures [1], any processing parameter that changes materials properties becomes critical. Sintering temperatures for conventional sintering of HAp are usually in the range from 850 to 1200 °C, while holding times can vary from 1 to even 20 h, depending on the sintering temperature and regime used [2-4]. Heating rate influences the thermal history during sintering process and forms the initial microstructure before foreseen isothermal sintering step. This can be an even stronger for materials that undergo structural chemical changes during sintering, which is the case for HAp ceramics that could lose structural carbonate and/or hydroxyl groups. Moreover, during sintering of non-stoichiometric HAp ceramics, change in phase composition might happen which additionally influences the microstructural evolution [2-4].

This study addresses non-isothermal densification behavior of HAp nanorods with heating rates from 2 to $50\,^{\circ}$ C/min up to $1200\,^{\circ}$ C, as well as corresponding microstructural and phase characterizations. It is showed that fully densified ceramics can be

obtained in significantly reduced time frame. The obtained results are important for further improving the both HAp synthesis and sintering pathways.

2. Materials and methods

HAp nanorods are synthesized by the chemical precipitation method in open reactor conditions, at ambient pressure. The initial ratio of Ca/P was set to stoichiometric 1.67 for HAp. Ca(NO₃)₂·4H₂O and NH₄H₂PO₄ were separately dissolved in distilled water, accompanied with the addition of NH₄OH to adjust pH value to 11. The solution of phosphates was dropwise added to calcium solution. The reactor temperature was 50 °C during the reaction time of 1.5 h. After 24 h in a native solution, the slurry was washed out to the neutral pH. Drying was performed for 18 h at 100 °C. XRD measurements (Rigaku Smartlab SAXS diffractometer, Japan) with the step of 0.05, and dwell time of 1.0 deg/min. Reflections of HAp are assigned referencing to ICSD card number # 99358. Particle size distribution (PSD) measurements (Malvern, UK) were performed from water dispersion of HAp powder. Scanning electron microscopy (SEM) imaging of HAp powder was done on Pt/Pdcoated samples (JEOL, Japan), while fractured surfaces of sintered samples were imaged on field-emission SEM (TESCAN, Czech Republic). Fourier transform infrared (FTIR) spectra of the samples were recorded at ambient conditions (Nicolet IS 50, USA), operating in the ATR mode and resolution of 4 cm⁻¹with 32 scans. The

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green samples were prepared by uniaxial pressing at 200 MPa, in 8 mm Ø mold. Non-isothermal sintering was performed in a vertical dilatometer system (Setsys Evolution, Setaram, France) with heating/cooling rates of 2, 10, and 50 °C/min up to 1200 °C with no holding time at that temperature. Sintering curves were calculated from the recorded change in sample's height, assuming isotropic shrinkage [5], which was experimentally confirmed by geometrical measurements after sintering.

3. Results and discussion

XRD pattern of synthesized material, Fig. 1a, shows the presence of reflections that correspond to the single phase HAp. The crystallite size in [0 0 2] crystallographic direction is estimated to 36 nm according to Scherrer's equation. FTIR spectrum, Fig. 1b, reveals the bands originating from phosphate and hydroxyl groups vibrations, as well as traces of incorporated carbonates into the crystal lattice of HAp. Phosphate vibration modes exist at: v1 at 962 cm⁻¹, v2 at 473 cm⁻¹, v3 at 1024 and 1092 cm⁻¹, and v4 at 560 and 602 cm⁻¹. The peaks from hydroxyl group are present at 630 and 3569 cm⁻¹[6]. Vibration bands of carbonates at 874 and 1419 cm⁻¹ originate from incorporated species at the position of a phosphate group (B-type substitution), while the peak at 1455 cm⁻¹ can be ascribed to carbonates at the position of a hydroxyl group (A-type substitution). The positions of peaks corresponding to structural carbonate ions imply that the prepared material can be regarded as B-type carbonated HAp, although some amount of carbonates at the place of the hydroxyl group cannot be excluded. The peaks at 1640 cm⁻¹ and between 3100 and 3400 cm⁻¹ indicate the presence of retained surface water. There are no any traces of residual nitrates or ammonium ions which can affect thermal and sintering behavior. Fig. 1c shows the particle size and the morphology of powder which consists of well-dispersed particles with a length of 300 nm and width of 30 nm. This finding is in a good agreement with the results of PSD, Fig. 4d, where the number particle size distribution indicates a peak around 350 nm. However, the presence of agglomerates of several micrometers in size is seen in volume PSD, which is a typical feature of nanopowders synthesized without the presence of agents that prevent agglomeration.

Non-isothermal sintering curves of HAp nanopowder are shown in Fig. 2a. As can be seen, the choice of heating rate influences the position of sintering curves, shifting them towards higher temperatures with the heating rate. While there is no significant difference between 2 and 10 °C/min, processing of HAp with a heating rate 50 °C/min shifts the intermediate sintering stage for almost 150 degrees, although sample achieves the similar final density at 1200 °C. After sintering, all samples have relative density >98% of theoretical density of HAp, based on geometrical measurements, which is usually considered as a fully dense material (>96%). XRD patterns of sintered samples presented in Fig. 2b show that the only present phase is pure HAp regardless the heating rate, which confirms the thermal stability of the material. Investigation of microstructural properties of non-isothermally sintered ceramics, Fig. 3, reveals that all samples possess practically fully dense microstructure. The average grain size is decreasing for the samples processed with faster heating: from micrometer-level in the case of 2 °C/min, 500 nm for 10 °C/min, and finally around 250 nm for the fastest heating rate of 50 °C/min. In the inset of

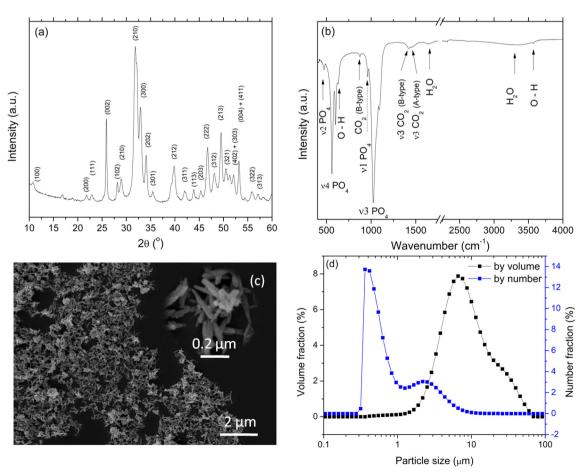


Fig. 1. Physicochemical characterization of prepared HAp: (a) XRD pattern, (b) FTIR spectrum, (c) FE-SEM image, and (d) PSDs.

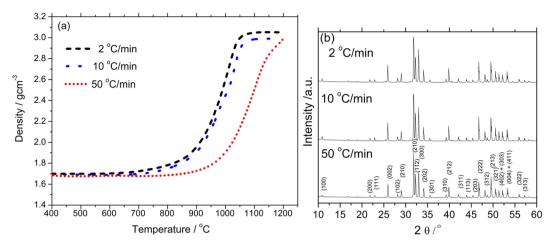


Fig. 2. (a) Non-isothermal sintering curves with different heating rates and (b) XRD patterns after sintering.

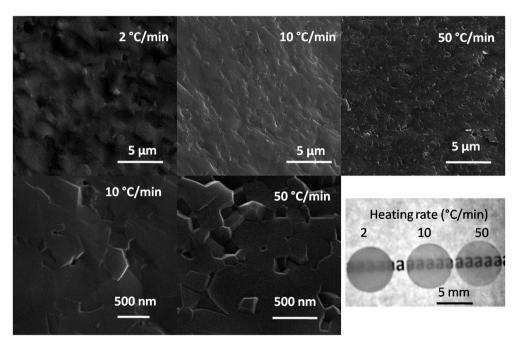


Fig. 3. SEM-SE (secondary electron) images of fractured surfaces of sintered ceramics; an inset shows the apparent translucency of 1 mm thick samples.

Fig. 3, a qualitative evidence of different optical translucency of sintered HAp specimens prepared with different heating rates is given. Such behavior was also observed for ZnS, however, the beneficial effect was found for the slowest heating rate of 5 °C/min [7].

FTIR spectra of sintered samples, Fig.4a, exhibit the presence of characteristic phosphate vibration modes, as well as that of a hydroxyl group. The bands from incorporated carbonates are almost diminished, especially the band at 874 cm⁻¹, while traces around 1456 cm⁻¹ can be discussed. This finding is in accordance with a recent study on sintered HAp-based ceramics [6]. The increase in the heating rate, and thus less time spent at higher temperatures, influences the relative amount of structural hydroxyl group, Fig.4b, where the intensity of hydroxyl vibration at 630 cm⁻¹is getting stronger with the increase in heating rate, implying lower structural dehydroxylation.

The presented results show that the choice of the appropriate heating rate is a very important step in sintering of HAp ceramics and that interplay of heating rate and sintering temperature has to be carefully chosen to synthesize materials with desired microstructural properties. Moreover, this is the first time that fully dense HAp is obtained by a single conventional heating step in less than 25 min, having an increased level of optical translucency. These findings could be explained by the beneficial alignment of HAp nanorods during the fast heating rate processing, achieved microstructural refinement, and higher amount of structural hydroxyl groups due to the significantly less time spent at high temperatures.

4. Conclusion

Variation of the heating rate during processing of HAp ceramics was successfully applied to densify HAp powdered material. The application of faster heating did not influence the final density of ceramics and contributed to the grain size refinement. The

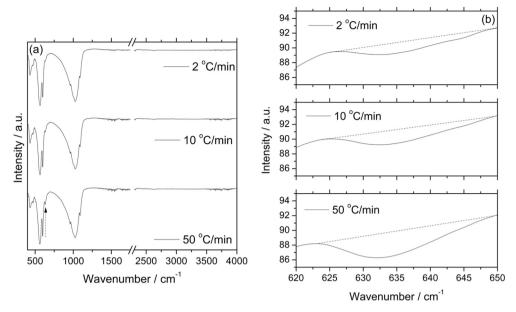


Fig. 4. (a) FTIR spectra after non-isothermal sintering and (b) the part of FTIR spectra in the range from 620 to 650 cm⁻¹ showing the structural O-H vibration band.

retention of a higher amount of structural hydroxyl groups was detected with the heating rate increase, as well as the optical translucency of sintered specimens. The obtained findings are useful standpoint for further research in sintering of HAp ceramics by application of an optimized heating rate. It is foreseen that such approach accompanied with the techniques suitable for microstructural optimization, like two-step sintering, can introduce an additional advantage since the driving force for densification would be preserved to a larger extent.

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